

Iwona Sulima*, Paweł Hyjek

University of the National Education Commission, Krakow
ul. Podchorążych 2, 30-084 Krakow, Poland

*Corresponding author: E-mail: iwona.sulima@up.krakow.pl

Received (Otrzymano) 23.05.2023

THE TRIBOLOGICAL CHARACTERISTICS OF ZIRCONIUM DIBORIDE PARTICLE REINFORCED COMPOSITES PROCESSED BY SPS

<https://doi.org/10.62753/ctp.2024.01.1.1>

The tribological properties of composites based on 316L steel reinforced with zirconium diboride (5 and 10 wt.%) were described. The composites were fabricated by the spark plasma sintering (SPS) method. Wear resistance tests were carried out using the ball-on-disc method at room temperature under dry friction conditions. Countersamples made of SiC, Si₃N₄, ZrO₂ and AISI52100 steel were used in the tests. The determined parameters included the coefficient of friction μ , mass loss Δm and specific wear rate $W_{s(disc)}$. After the tribological tests, the surfaces of the samples were examined under a confocal microscope. The obtained research results showed that the tribological properties depend on the test conditions (the type of countersample) and on the content of the ZrB₂ reinforcing phase.

Keywords: composites, zirconium diboride (ZrB₂), ball-on-disc method, wear rate, coefficient of friction

CHARAKTERYSTYKA TRIBOLOGICZNA KOMPOZYTÓW WZMOCNIONYCH CZĄSTKAMI DIBORKU CYRконU SPIEKANYCH METODĄ SPS

Przedstawiono właściwości tribologiczne kompozytów na osnowie stali 316L wzmocnionych diborkiem cyrkonu (5 i 10% wag.). Kompozyty zostały wytworzone z zastosowaniem metody Spark Plasma Sintering (SPS). Badania odporności na zużycie przeprowadzono, wykorzystując metodę ball-on-disc w temperaturze pokojowej w warunkach tarcia suchego. Testy przeprowadzono, stosując następujące przeciwpróbki: SiC, Si₃N₄, ZrO₂ i stal AISI52100. Wyznaczono współczynnik tarcia μ , ubytek masy Δm i wskaźnik zużycia $W_{s(disc)}$. Powierzchnie próbek po testach tribologicznych obserwowano za pomocą mikroskopu konfokalnego. Uzyskane wyniki badań wykazały, że właściwości tribologiczne zależą od warunków badań (rodzaju przeciwpróbki) i ilość fazy wzmacniającej ZrB₂.

Słowa kluczowe: kompozyty, diborek cyrkonu (ZrB₂), metoda ball-on-disc, szybkość zużycia, współczynnik tarcia

INTRODUCTION

Abrasive wear is the dominant type of wear under conditions of dry friction, and this process is related to the bulk properties of the material. Abrasive wear is a mechanical process of material wear [1, 2]. The resistance to abrasive wear depends on the test conditions (friction force, rotational speed, distance, temperature) and on the type of tribological pair [3, 4]. The wear behaviour of composite materials is affected by the morphology, type, amount, arrangement and properties of the reinforcing phase [5, 6]. According to literature [7-10], studies were carried out to explain the effect of various ceramic particles (TiB₂, TiC, Al₂O₃, Y₂O₃, SiC, MoS₂) on the tribological properties of composites based on stainless steel. The introduction of a ceramic phase into the steel matrix improves the abrasion resistance of such materials [11-14]. Tjong and Lau [8] investigated the effect of changing load (15, 35 and 55

N) and sliding speed (1 and 3 m s⁻¹) on the wear behaviour of 304 steel+20 wt.% TiB₂ composites. It was shown that the wear rate of the 304 steel without the reinforcing phase grows rapidly with increasing load and sliding speed. On the other hand, the opposite trend was observed in the 304 steel+20 wt.% TiB₂ composites, where the volumetric wear decreased with the increase in the applied load and sliding speed. Srivastava and Das [15] demonstrated that composites reinforced with TiC and (Ti, W)C offer better wear resistance than austenitic steel without reinforcement. The research demonstrated the superior resistance to abrasive wear of the composites containing (Ti, W)C as compared to the composites reinforced with TiC. It was also been shown that the abrasive wear resistance and coefficient of friction of austenitic steel and composites reinforced with particles of TiC and (Ti, W)C decrease with increasing load.

The main purpose of this study is to investigate the tribological properties of steel+5% ZrB₂ and steel+10% ZrB₂ composites. The effect of the different countersample type and ZrB₂ on the coefficient of friction, mass loss and specific wear rate was studied.

MATERIALS AND METHODS

In the present work, powders of 316L austenitic stainless steel (particle size of 25 μm, Hogan), zirconium diboride (ZrB₂, particle size of 2.5-3.5 μm, H.C. Starck) were used as the raw material for the research. ZrB₂ particles were added to the steel powders in amounts of 5 and 10 weight percent. The composites were sintered using spark plasma sintering (SPS) in vacuum under the pressure of 35 MPa, at the temperature of 1373 K and for the time of 300 seconds. The density of the sintered composites was determined by the Archimedes' method [16]. Young's modulus was measured using an ultrasonic flaw detector ZBM ULTRA. The microstructure of the sintered materials was evaluated by means of a scanning electron microscope (SEM) Hitachi SU-70 (Tokyo, Japan) with wavelength dispersive spectroscopy (WDS). Next, the hardness was determined using the microhardness tester NEXUS 4000 (Innovatest).

Vickers hardness measurements were carried out with a 2.942 N load and a loading time of 10 s. The tests were conducted in accordance with the requirements of the Polish Standard [17].

The abrasion resistance tests of the sintered steel+ZrB₂ composites were performed by the ball-on-disc method using an ELBIT universal tribotester in accordance with the standard [18]. All the tests were carried out under the following conditions: dry friction, temperature of 22 °C, load of 5 N, speed of 0.1 m/s and time of 10,000 seconds. Countersamples (balls with a diameter of 3.175 mm) made of various materials, i.e. SiC, Si₃N₄, ZrO₂ and AISI52100 steel, were used in the tests. During the test, each sample was paired with a new ball surface [19]. Before the tests, the balls and the samples were washed with ethyl alcohol in an ultrasonic washer. They were weighed on a RADWAG AS 220/C/2 balance before and after each abrasion test.

Mass loss (Δm) was calculated from the following equation:

$$\Delta m = \frac{m_0 - m_k}{m_0} 100\% \quad (1)$$

where: m_0 – initial mass [g], m_k – final mass [g].

The specific wear rate was calculated by means of the equation:

$$W_{V(disc)} = \frac{V_{disc}}{F_n \cdot L} \quad (2)$$

where: F_n – applied load [N], V_{disc} – wear volume of the disc specimen [mm³], L – sliding distance [m].

The wear volume of the disc specimen was calculated from following equation:

$$V_{disc} = \frac{\pi}{2} \cdot R \cdot (S_1 + S_2 + S_3 + S_4) \quad (3)$$

where: R – radius of wear track [mm], S_1 to S_4 – cross-sectional areas at four places on the wear track circle [mm²] [19].

When the tests were completed, the abrasion marks were subjected to microscopic analysis employing an OLYMPUS LEXT OLS5100 confocal microscope with OLS5100 Analysis Application software dedicated to surface analysis.

RESULTS AND DISCUSSION

SEM micrographs of the steel+ZrB₂ composites are shown in Figures 1-3. The WDS examinations of the chemical composition confirmed the presence of ZrB₂ particles (bright phase) evenly distributed along the grain boundaries in the steel matrix. The matrix was also observed to contain very fine (below 1 μm) precipitates of chromium.

The physical and mechanical properties are presented in Table 1. For the steel+ZrB₂ composites sintered by the SPS method, a high degree of compaction was obtained using the sintering time of 300 seconds. The apparent density of the sinters was in the range of 94-95% of the theoretical density. The results showed that the apparent density gradually decreased with increasing content of ZrB₂ in the steel matrix. This effect was due to the lower density of zirconium diboride (6.08 g/cm³) compared to the density of 316L steel (8.00 g/cm³) [20, 21].

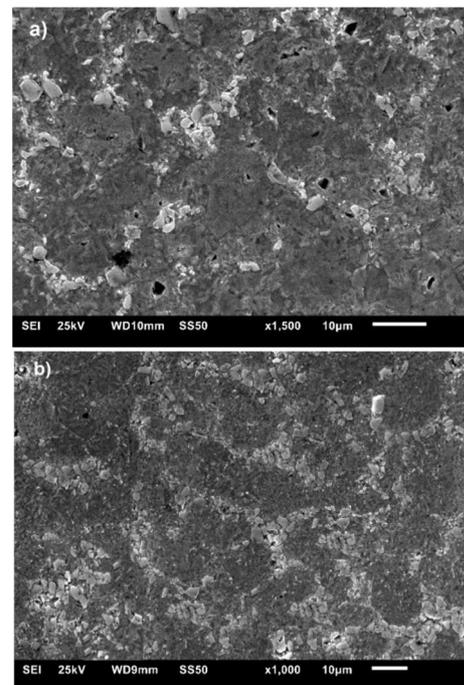


Fig. 1. SEM micrographs showing distribution of ZrB₂ reinforcement in: a) steel+5% ZrB₂, b) steel+10% ZrB₂ composites

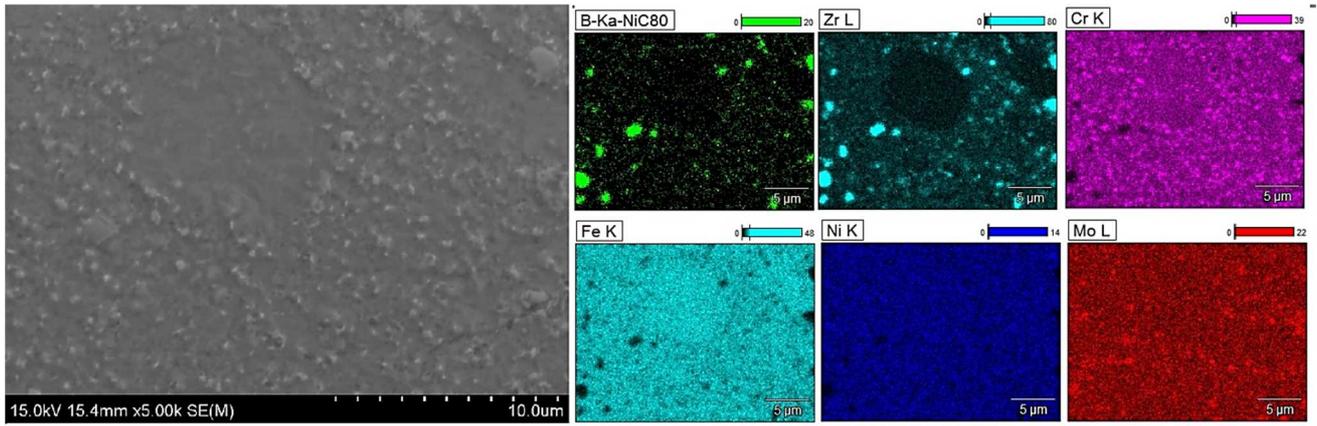


Fig. 2. Microstructure (SEM) and WDS composition analysis of steel+5% ZrB₂ composite

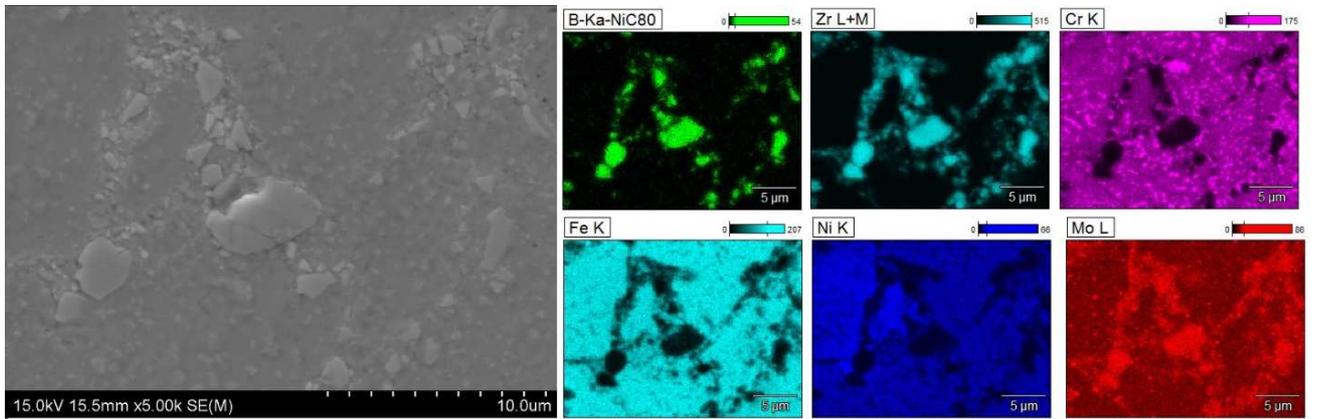


Fig. 3. Microstructure (SEM) and WDS composition analysis of steel+10% ZrB₂ composite

TABLE 1. Properties of steel+ZrB₂ composites sintered by SPS method

Sintered composites	Relative density [g/cm ³]	Apparent density [%]	Porosity [%]	Young's modulus [GPa]	Hardness [HV _{0.3}]
steel	7.82	98	0.58	197	212
steel+5% ZrB ₂	7.35	94	0.71	210	297
steel+10% ZrB ₂	7.22	95	0.82	218	381

The composite sinters were characterized by a low open porosity in the range of 0.71-0.82%. The ultrasonic method was used to determine Young's modulus. The value of Young's modulus depends on the microstructure of the sintered composite. The presence of a large number of defects and inhomogeneities in the material affects the attenuation of ultrasonic waves during measurements and low values of Young's modulus are obtained [22, 23]. Data analysis (Table 1) showed that with an increasing content of the reinforcing phase, Young's modulus also increased and amounted to 205 GPa and 216 GPa for the composites reinforced with 5% and 10% ZrB₂, respectively. According to literature [24], Young's modulus of 316L steel sintered under the same conditions is 196 GPa. The research (Table 1) showed that the hardness of the composites also gradually improved with the increasing addition of ZrB₂ particles. For the composites containing 10% ZrB₂

it was 381 HV_{0.3} against only 225 HV_{0.3} [25] obtained for the sintered 316L steel.

Using the ball-on-disc method, the tribological properties of the steel+ZrB₂ composites were investigated. Figure 4 shows changes in the coefficient of friction depending on the test time. From the obtained results (Table 2) it follows that the tribological properties of the sintered composites depend on the content of the reinforcing phase and on the type of countersample used. By examining the effect of the reinforcing phase, it can be concluded that the increase in the content of ZrB₂ in the steel matrix causes an increase in the value of the coefficient of friction and a simultaneous decrease in the mass loss and wear index (Table 2, Fig. 5). This trend prevailed in all the tests performed with different countersamples. Additionally, the analysis of the results shows that the coefficient of friction depends on the type of countersample used.

The highest values of the coefficient of friction, i.e. 0.57 and 0.62, were obtained for the steel+5% ZrB₂ and steel+10% ZrB₂ composites, respectively, using AISI52100 steel countersamples during the tests. The use of ceramic countersamples significantly reduced the coefficient of friction of the composites. The lowest coefficient was obtained during tests with the ZrO₂ countersample.

Introducing ZrB₂ particles in the amounts of 5 and 10 wt.% into the steel matrix gradually reduced the val-

ue of the wear rate (Table 2, Fig. 5b). With a higher content of the reinforcing phase, a lower value of the wear rate was obtained. This effect was clearly visible in the composites tested with ceramic countersamples, i.e. Si₃N₄, SiC, ZrO₂. It is very important that the wear rate assumed different values depending on the type of counter-sample used. The lowest value of the wear rate ($42 \cdot 10^{-6}$ mm³/Nm) was obtained for the steel+10% ZrB₂ composites tested with the ZrO₂ counter-sample.

TABLE 2. Tribological properties of composites sintered by SPS method

Sintered composites	Coefficient of friction μ [-]	Relative mass loss Δm [%]	Specific wear rate $\cdot 10^{-6}$ W_v [mm ² /Nm]
AISI52100 steel			
steel	0.67	0.125	319
steel+5% ZrB ₂	0.57	0.092	221
steel+10% ZrB ₂	0.62	0.071	205
ball of Si ₃ N ₄			
steel	0.61	0.158	456
steel+5% ZrB ₂	0.40	0.068	119
steel+10% ZrB ₂	0.46	0.031	64
ball of SiC			
steel	0.59	0.189	516
steel+5% ZrB ₂	0.35	0.063	132
steel+10% ZrB ₂	0.48	0.021	81
ball of ZrO ₂			
steel	0.57	0.205	610
steel+5% ZrB ₂	0.30	0.029	112
steel+10% ZrB ₂	0.47	0.015	42

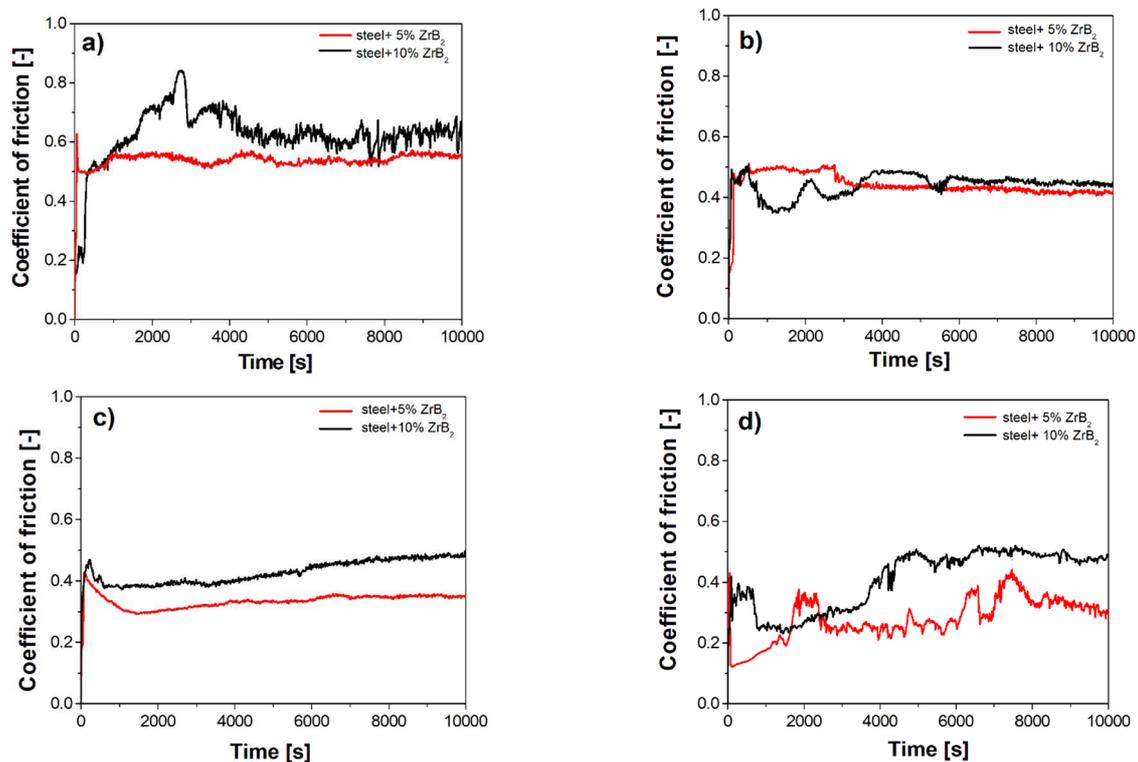


Fig. 4. Coefficient of friction of steel+5% ZrB₂ and steel+10% ZrB₂ composites as function of testing time, measured using different countersamples of: a) AISI52100 steel, b) Si₃N₄, c) SiC and d) ZrO₂

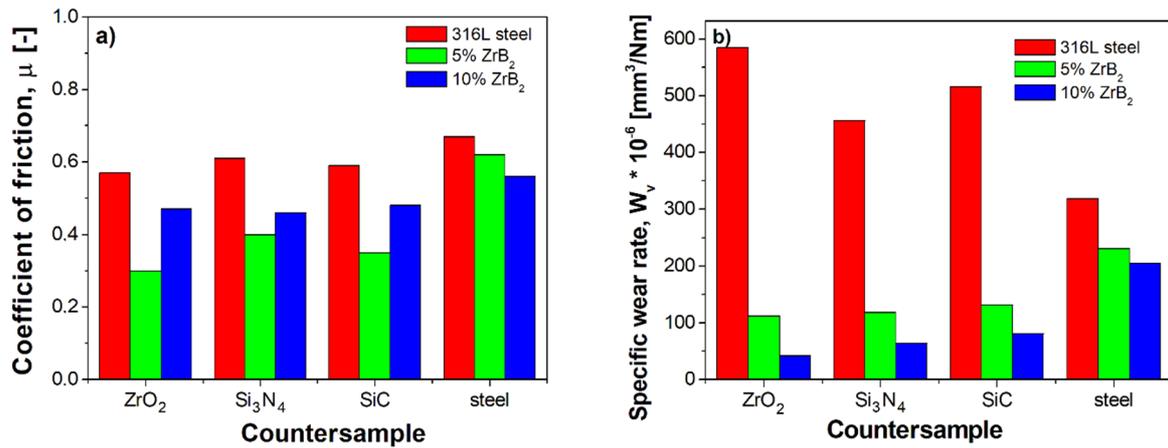


Fig. 5 Coefficient of friction (a) and specific wear rate measured using different countersamples (b)

After the tribological tests, microscopic analysis and geometrical measurements of the abrasion area were carried out for each sample using a confocal microscope.

Figures 6 and 7 include micrographs of the microstructures showing the surface condition and wear tracks left by the abrasion tests. Figure 8 compares examples of the wear profiles at four different spots in the steel+10% ZrB₂ composite sample after tests using a ZrO₂ counter-sample. The analysis of the wear tracks revealed that in all the tested composites, abrasion was the leading mechanism responsible for the wear.

Scratches, damage and pits were observed in the area of the worn surfaces (abrasive type of wear). Moreover, local accumulation of wearing edges was observed in the researched materials (Fig. 8), which also indicates the abrasive nature of the wear.

Figure 9 presents the SEM micrographs of an exemplary surface of the steel+ZrB₂ composites after tests using a ZrO₂ countersample. Abrasive wear is present in the all composites as the scratches in the wear track indicate. Plastic deformation with characteristic grooves was observed.

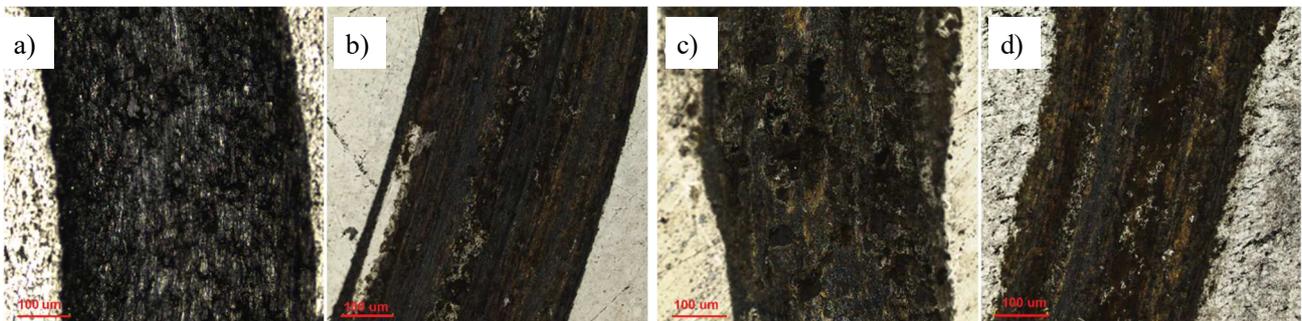


Fig. 6. Micrographs of worn surface of steel+5% ZrB₂ composites after wear test using different countersamples of: a) AISI52100 steel, b) SiC, c) Si₃N₄ and d) ZrO₂

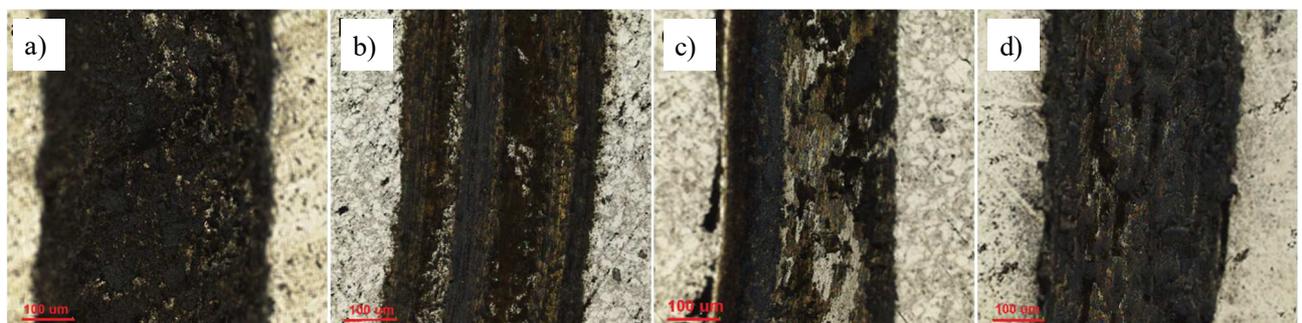


Fig. 7. Micrographs of worn surface of steel+10% ZrB₂ composites after wear test using different countersamples of: a) AISI52100 steel, b) SiC, c) Si₃N₄ and d) ZrO₂

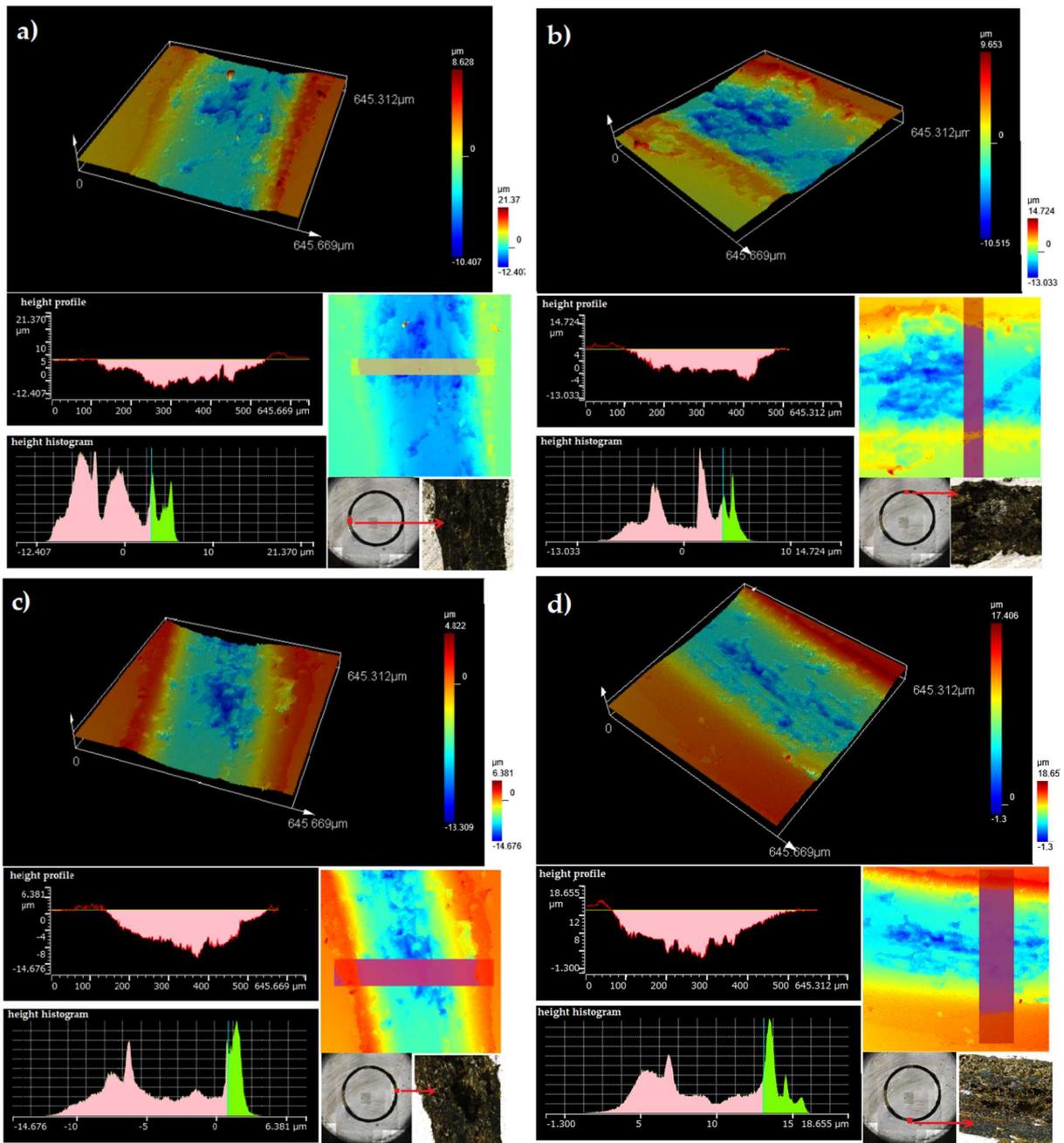


Fig. 8. Comparison of results of analysis of wear tracks and wear profiles of steel+10% ZrB₂ composite sample examined under Olympus LEXT OLS5100 confocal microscope (ZrO₂ countersample)

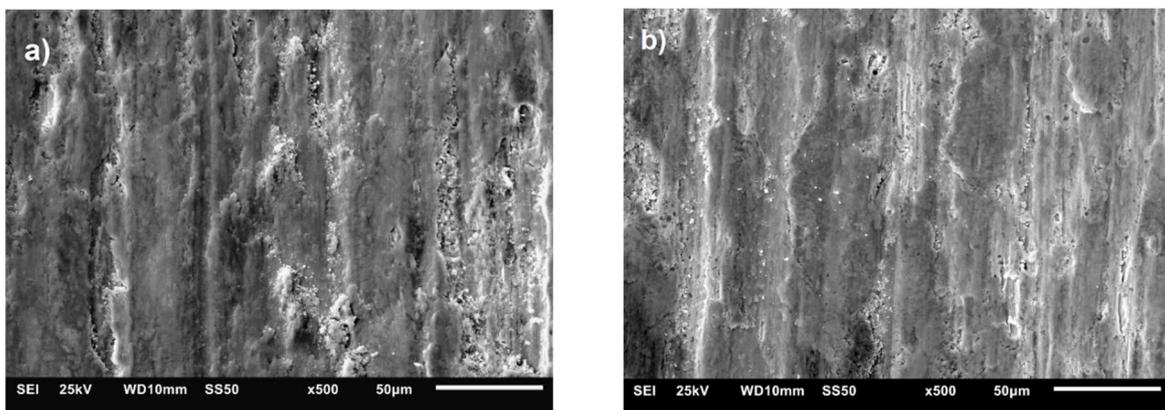


Fig. 9. SEM micrograph of worn surface of steel+10% ZrB₂ (a) and steel+20% ZrB₂ (b) composite after tests using ZrO₂ countersample

CONCLUSIONS

The increase in the mass fraction of ZrB₂ increases the density, Young's modulus and hardness of the 316L steel-based composites sintered by the SPS method.

The results of the wear test have shown that increasing the content of the ZrB₂ reinforcing phase improves the wear resistance of the composites. The wear rate depends on the type of countersample used. Among all the tested friction pairs, the lowest value of wear rate was obtained for the steel+10% ZrB₂ composite tested with the countersample made of ZrO₂.

The coefficient of friction rises with the increasing content of ZrB₂ when the tests are conducted with the ceramic countersamples made of Si₃N₄, SiC, and ZrO₂. The lowest value of the coefficient of friction was obtained for the steel+5% ZrB₂ composite tested with the countersample made of ZrO₂.

Funding

This research was funded by statutory funds of the Institute of Technology, Pedagogical University of Cracow grant number WPBU/2023/03/00001. The project for the purchase of scientific and research equipment: "Innovative research and scientific platform for a new class of nanocomposites", was financed by the Ministry of Education and Science, contract number 7216/IA/SP/2021.

REFERENCES

- [1] Gierek A., Zużycie tribologiczne, Wydawnictwo Politechniki Śląskiej, Gliwice 2006 (in Polish).
- [2] Hebda M., Wachal A., Trybologia, Wydawnictwo Naukowo-Techniczne, Warszawa 1980 (in Polish).
- [3] Płaza S., Margielewski L., Celichowski G., Wstęp do tribologii i tribochemia, Wydawnictwo Uniwersytetu Łódzkiego, Łódź 2005 (in Polish).
- [4] Velasco F., Gordo E., Isabel R., Ruiz-Navas E.M., Bautista A., Torralba J.M., Mechanical and wear behaviour of high-speed steels reinforced with TiCN particles, International Journal of Refractory Metals & Hard Materials 2001, 19, 319-323, DOI: 10.1016/S0263-4368(01)00053-1.
- [5] Vardavoulias M., Jouanny-Tresy C., Jeandin M., Sliding-wear behaviour of ceramic particle-reinforced high-speed steel obtained by powder metallurgy, Wear 1993, 165, 2, 141-149, DOI: 10.1016/0043-1648(93)90329-K.
- [6] Akhtar F., Microstructure evolution and wear properties of in situ synthesized TiB₂ and TiC reinforced steel matrix composites, Journal of Alloys and Compounds 2008, 459, 491-497, DOI: 10.1016/j.jallcom.2007.05.018.
- [7] Vardavoulias M., Jeandin M., Velasco F., Torralba M., Dry sliding wear mechanism for P/M austenitic stainless steel and their composites containing Al₂O₃ and Y₂O₃ particles, Tribology International 1998, 28, 6, 499-506, DOI: 10.1016/0301-679X(95)00110-P.
- [8] Tjong S.C., Lau K.C., Tribological behaviour of SiC particle-reinforced copper matrix composites, Materials Letters 2000, 43, 5-6, 274-280, DOI: 10.1016/S0167-577X(99)00273-6.
- [9] Surapol R., Sithipong M., Ruangdaj T., Tribological behaviour of sintered 316L stainless steel impregnated with MoS₂ plain bearing, Wear 2008, 265, 3-4, 546-553, DOI: 10.1016/j.wear.2007.11.014.
- [10] Niranjana K., Lakshminarayanan P.R., Dry sliding wear behaviour of in situ Al-TiB₂ composites, Materials and Design 2013, 4, 167-173, DOI: 10.1016/j.matdes.2012.11.035.
- [11] Raadnuj S., Mahathananabodee S., Tong Sri R., Tribological behaviour of sintered 316L stainless steel impregnated with MoS₂ plain bearing, Wear 2008, 265, 546-553, DOI: 10.1016/j.wear.2007.11.014.
- [12] Tjong S.C., Lau K.C., Sliding wear of stainless steel matrix composite reinforced with TiB₂ particles, Materials Letters 1999, 41, 153-158, DOI: 10.1016/S0167-577X(99)00123-8.
- [13] Sulima I., Tribological properties of steel/TiB₂ composites prepared by spark plasma sintering, Archives of Metallurgy and Materials 2014, 59, 4, 1263-1268, DOI: 10.2478/amm-2014-0216.
- [14] Zhang Z., Chen Y., Zhang Y., Gao K., Zuo L., Qi Y., Tribology characteristics of ex-situ and in-situ tungsten carbide particles reinforced iron matrix composites produced by spark plasma sintering, Journal of Alloys and Compounds 2017, 704, 260-268, DOI: 10.1016/j.jallcom.2017.02.003.
- [15] Srivastava A.K., Das K., The abrasive wear resistance of TiC and (Ti,W)C-reinforced Fe-17Mn austenitic steel matrix composites, Tribology International 2010, 43, 5-6, 944-950, DOI: 10.1016/j.triboint.2009.12.057.
- [16] Taylor R.P., McClain S.T., Berry J.T., Uncertainty analysis of metal-casting porosity measurements using Archimedes' principle, International Journal of Cast Metals Research 1999, 11, 247-257, DOI: 10.1080/13640461.1999.11819281.
- [17] PN-EN ISO 6507-1:2018-05 – Metale – Pomiar twardości sposobem Vickersa – Część 1, Metoda badania (in Polish).
- [18] International Standard, Fine ceramics (advanced ceramics, advanced technical ceramics) Determination of friction and wear characteristics of monolithic ceramics by ball-on-disc method, ISO 20808:2004(E).
- [19] Meozzi M., Special use of the ball on disc standard test, Tribology International 2006, 39, 6, 496-505, DOI: 10.1016/j.triboint.2005.03.011.
- [20] Cutler R.A., Engineering Properties of Borides, Ceramics and Glasses: Engineered Materials Handbook, ASM International, Materials Park, OH, 1991.
- [21] McGuire M.F., Stainless Steels for Design Engineers, Austenitic Stainless Steel, Chapter 6, ASM International 2008.
- [22] Fan Z., Tsakiroopoulos P., Miodownik A.P., Prediction of Young's modulus of particulate two phase composites, Materials Science and Technology 1992, 8, 10, 922-929, DOI: 10.1179/mst.1992.8.10.922.
- [23] ASTM E1876-09, Standard Test Method for Dynamic Young's Modulus, Shear Modulus, and Poisson's Ratio by Impulse Excitation of Vibration 2009, 15.
- [24] Sulima I., Role of boron addition on the consolidation and properties of steel composites prepared by SPS, Bulletin of Materials Science 2015, 38, 1831-1841, DOI: 10.1007/s12034-015-0984-y.
- [25] Sulima I., Putyra P., Hyjek P., Tokarski T., Effect of SPS parameters on densification and properties of steel matrix composites, Advanced Powder Technology 2015, 26, 4, 1152-1161, DOI: 10.1016/j.apt.2015.05.010.